

# Chitosan-Dextran Sulfate Hydrogels as a Potential Carrier for Probiotics

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## ABSTRACT

Physical and chemical (crosslinked with genipin) hydrogels based on chitosan and dextran sulfate were developed and characterized as novel bio-materials suitable for probiotic encapsulation. The swelling of the hydrogels was dependent on the composition and weakly influenced by the pH of the media. The morphology analysis supports the swelling data showing distinct changes in microstructure depending on the composition. The viability and culturability tests showed approx. 3.6 log CFU/mL decrease of cells (*L. acidophilus* as model) incorporated into chemical hydrogels when compared to the number of viable native cells. However, the live/dead viability assay evidenced that a considerable amount of viable cells were still

19 entrapped in the hydrogel network and therefore the viability is most likely underestimated. Overall, the  
20 developed systems are robust and their structure, rheology and swelling properties can be tuned by  
21 changing the blend ratio, thus constituting appealing bio-matrices for cell encapsulation.

22 Keywords: Chitosan; dextran sulfate; genipin; hydrogel; encapsulation; probiotic bacteria.

## 23 **1 Introduction**

24 Hydrogels are typically defined as tridimensional polymer networks that can take up considerable  
25 amounts of solvent without dissolving due to its inherent hydrophilicity (Bhattacharai, Gunn, & Zhang, 2010).  
26 Hydrogels can be used to form different physical structures such as microparticles or nanoparticles, and  
27 coatings (Hoare & Kohane, 2008). They are often highly deformable, mainly due to their unique structural  
28 tunability. Their porous structure and, consequently, swelling performance can be altered by varying the  
29 crosslinking density within the gel matrix (Hoare & Kohane, 2008). Moreover, these systems can respond to  
30 external stimuli, such as pH, light, electric field, and thus change volume and shape (Shang, Shao, & Chen,  
31 2008). Hydrogels are therefore suitable for a broad range of applications such as drug delivery (Gupta,  
32 Vermani, & Garg, 2002; Saboktakin, Tabatabaie, Maharramov, & Ramazanov, 2010), tissue engineering  
33 (Drury & Mooney, 2003; Lee & Mooney, 2001), biomedicine (Berger, Reist, Mayer, Felt, & Gurny, 2004;  
34 Shang et al., 2008), and encapsulation technologies (Altunbas, Lee, Rajasekaran, Schneider, & Pochan,  
35 2011; Karoubi, Ormiston, Stewart, & Courtman, 2009).

36 Currently, there is an increasing need for replacing synthetic hydrogel based products by natural  
37 counterparts prepared with sustainable materials from renewable resources mainly due to stricter  
38 international regulations. In this context, chitosan (CH) emerges as a popular cationic biopolymer for the  
39 formulation of colloidal hydrogels. CH is a random linear copolymer of (1-4)-N-acetyl-D-glucosamine and (1-  
40 4)-D-glucosamine units, produced by the deacetylation of the naturally occurring chitin under high alkaline  
41 conditions (Vårum & Smidsrød, 2005). Its polycationic feature is due to amine residues that are protonated

42 below pH ca. 6.5 (Illum & Davis, 2005). Moreover, CH is biocompatible, non-toxic (Ravi Kumar, 2000), and  
43 biodegradable (i.e. it can be metabolized by enzymes, such as lysozyme, present in human body fluids)  
44 (Prabaharan & Mano, 2005) and potentially by bacterial enzymes in the colon (Kean & Thanou, 2010). On  
45 the other hand, dextran sulfate (DXS) is one of the most studied anionic, biocompatible, and biodegradable  
46 biopolymers. It is a branched polysaccharide with 1–6 and 1–4 glycosidic linkage and contains  
47 approximately 16-19 % sulfur, which is equivalent to approximately 1.6 - 2.4 sulfate groups per glucosyl unit  
48 (Anitha et al., 2011). It is widely used in food, materials and pharmaceutical areas. Hence, DXS and CH are  
49 promising candidates for the encapsulation of different cargo due to their safety and bio-profiles. Apart  
50 from the physical systems, CH can be chemically crosslinked with remarkable impact on the rheological and  
51 physicochemical properties of formed hydrogels (Delmar & Bianco-Peled, 2015; Dimida et al., 2015). The  
52 formation of permanent covalent bonding between the biopolymer chains also improves the mechanical  
53 properties in comparison to the related physical hydrogels. In particular, genipin (GP, isolated from the  
54 gardenia plant) is an ideal crosslinker agent due to its biocompatibility and considerably lower toxic profile  
55 than glutaraldehyde, formaldehyde or other often used crosslinker agents (Bhattacharai et al., 2010; Dimida et  
56 al., 2015). GP can react with the primary amine group of CH to create stiffer and stronger hydrogel network  
57 (Bhattacharai et al., 2010). In addition, geniposide, abundant in some fruits, was reported to be transformed to  
58 GP in the gastrointestinal tract without an adverse effect (Butler, Ng, & Pudney, 2003).

59 Although the formation of physical CH-DXS hydrogels was reported earlier (Delair, 2011), we further  
60 explored the formation of GP-crosslinked CH-DXS based hydrogels with focus not only on their preparation  
61 and characterization, but also on their performance to entrap probiotic cells. **Crosslinking can contribute to  
62 improve both the mechanical and pH stability of the hydrogel matrix during the harsh gastric passage.**  
63 Ultimately, the goal is to produce a responsive biomaterial delivery system for probiotic bacteria that  
64 ensures a safe passage through the harsh conditions found in the stomach while allowing the release of the  
65 encapsulated content at the human colon. As a proof of concept, model probiotic bacteria, *Lactobacillus*

66 *acidophilus*, was chosen to be incorporated into the physical and chemical hydrogels whereupon their  
67 viability was assessed.

## 68 **2 Materials and methods**

### 69 **2.1 Materials and chemicals**

70 CH (extracted and/or purified from *Pandalus borealis* shell, low molecular weight, deacetylation  $\geq 75\%$ ),  
71 acetic acid (glacial,  $\geq 99.85\%$ ), sodium chloride (NaCl, Fluka) and potassium dihydrogen phosphate ( $\text{KH}_2\text{PO}_4$ )  
72 were purchased from Sigma-Aldrich, Steinheim, Germany. DXS (from molecular weight 500 kDa, ultra-pure  
73 grade) was obtained from Amresco, Ohio, USA. The most relevant properties of the biopolymers used are  
74 summarized in Table S1 in supporting information, SI. GP was bought from Challenge Bioproducts, Taiwan.  
75 Disodium hydrogen phosphate dihydrate ( $\text{Na}_2\text{HPO}_4$ ) and potassium chloride (KCl) were obtained from  
76 Merck, Darmstadt, Germany. *Lactobacillus acidophilus* (LA5) was a kind gift from Chr. Hansen A/S,  
77 Copenhagen, Denmark. MRS broth (de Man, Rogosa and Sharpe), MRS agar, and atmosphere generation  
78 system (AnaeroGen sachets) were bought from Oxoid, Basingstoke, England. The MRS agar and broth were  
79 sterilized in an autoclave (115 °C, 10 min). The live/dead<sup>®</sup> BacLight™ bacterial viability kit for microscopy  
80 was bought from ThermoFisher Scientific, Molecular Probes, Eugene, OR, USA. All the chemicals were used  
81 as received. Milli-Q water (18.2 M $\Omega$ ·cm at 25 °C) was used in all experiments.

### 82 **2.2 Preparation of the biopolymer solutions**

83 Concentrated CH stock solutions (2, 3, and 4 wt %) were prepared by dissolving a known amount of CH  
84 into 1 % v/v glacial acetic acid. The solutions were stirred with a magnetic stirrer for at least 12 h until a  
85 clear solution was obtained. The DXS stock solutions (2 and 3 wt %) were simply prepared in Milli-Q water.  
86 Depending on the solution, all the necessary dilutions were made either with 1 % v/v glacial acetic acid or  
87 Milli-Q water without pH adjustment. The pH of 3 wt % CH was approx. 4.8, while it was approx. 7.0 for the  
88 2 wt % DXS.

### 89 **2.3 Preparation of physical and chemical hydrogels**

90 After the preparation of the pure biopolymer solutions, several samples were prepared by adding equal  
91 volume of the CH aqueous solution (typically 2.5 mL) into the DXS aqueous solution (typically 2.5 mL). The  
92 samples were stirred at room temperature until a homogeneous mixture was obtained (approx. 19-24 h).  
93 **The hydrogel samples were visually checked (naked eye) for several days.** To form the chemical hydrogels,  
94 GP stock solutions (0.5 and 1 wt %) were prepared in Milli-Q water and diluted when necessary. The most  
95 promising CH and DXS blend ratios were chosen to be crosslinked with different amounts of GP (from 0.05  
96 to 0.3 wt % in the final mixture). Note that the GP solution was added to the DXS solution prior to adding  
97 the CH solution. The final mixtures were stirred for 19 to 24 h at room temperature until homogeneous gels  
98 were obtained. **The compositions of hydrogels given in the following text refer to the component amount**  
99 **in the final mixtures.** The pH of final mixtures was between 4.7 – 4.9 (well below the pKa of CH, approx. 6.5)  
100 and therefore CH is expected to have a reasonably high charge density.

### 101 **2.4 Bead formation**

102 Apart from the bulky hydrogels, the formation of beads was explored by dropwise addition of different  
103 concentrations of the CH aqueous solutions (using a 200  $\mu$ L pipette tip) into different DXS solutions and vice  
104 versa. For the formation of the chemical crosslinked beads, the CH solution was dripped into DXS solution  
105 in which an appropriate amount of GP was previously mixed. The beads were left to cure at room  
106 temperature for 12 h.

### 107 **2.5 Rheological characterization**

108 The rheological properties of physical and chemical hydrogels were measured at 25 °C using Haake Mars  
109 III rheometer (Thermo Scientific, Karlsruhe, Germany) equipped with a cone and plate measuring geometry  
110 (1° cone angle, 35 mm diameter, and 0.05 mm gap). For the steady state viscosity measurements, the shear  
111 rate was linearly increased from 0.01 to 100  $s^{-1}$ . For the dynamic viscoelastic measurements, the linear  
112 viscoelastic regime was initially determined via stress sweep experiments (0.1 to 50 Pa) at constant

113 frequency (1 Hz). Frequency sweep test (from 10 to 0.01 Hz) was then conducted by applying a constant  
114 stress within the linear region (typically, 1 and 4 Pa for the physical and chemical hydrogels, respectively).  
115 The dynamic mechanical spectra were obtained recording the storage modulus ( $G'$ ) and loss modulus ( $G''$ )  
116 as a function of frequency.

## 117 **2.6 Swelling experiments**

118 The different physical and chemical hydrogels were tested regarding their swelling profile in aqueous  
119 solutions at pH 2.0 (0.1 M HCl) and pH 7.4 (0.1 M NaOH). The systems tested comprised hydrogels where  
120 the CH and GP concentrations were kept constant (1.5 and 0.1 wt %, respectively) while the DXS  
121 concentration was varied (i.e. 0, 1, and 1.25 wt %). Alternatively, CH and DXS concentrations were kept  
122 constant at 1.5 and 1 wt %, respectively, while the GP concentration was varied (i.e. 0, 0.1, and 0.15 wt %).  
123 Prior to the swelling experiments, the hydrogels were left to cure for a similar period (approx. 27-29 h). The  
124 swelling experiments were performed following an adapted method previously reported (Delmar &  
125 Bianco-Peled, 2015). In brief, the hydrogels (in triplicates, approx. 50-60 mg) were submerged in Eppendorf  
126 tubes containing 1.5 mL of aqueous solution at either pH 2.0 or 7.4. The tests were performed using a  
127 series of Eppendorf tubes, one for each sampling point (ie. 1, 2, 3, 4, 5, 10, 20, 30 min). Water evaporation  
128 was minimized by closing the Eppendorf lids during the experiment. Each gel was weighed periodically  
129 after removing the swelling media from the tubes. The swelling percentage,  $Q$  %, was determined over time  
130 using Eq.1.

$$131 \quad Q, \% = \frac{m_t - m_0}{m_0} \times 100 \quad (1)$$

132 where  $m_0$  is the initial weight of the hydrogel and  $m_t$  is the weight of the hydrogel at time  $t$ . Data were  
133 expressed by the average of  $Q$  % values from triplicates  $\pm$  standard deviation. The equilibrium swelling ( $Q_{eq}$   
134 %) was estimated from the  $Q$  % vs.  $t$  curve for each type of hydrogel upon reaching steady state.

## 135 **2.7 Morphological characterization by scanning electron microscopy**

136 The morphology of the hydrogels and beads was assessed by scanning electron microscopy (SEM, Zeiss  
137 EVO LS10) equipped with a LaB6 filament operating at 15 kV and with 50 pA probe current in high vacuum  
138 mode. Sample preparation was adapted from a previous method (Wang & Chen, 2016). In brief, the  
139 hydrogel samples or beads were spread and/or fixed on carbon tape on an aluminum stub and then frozen  
140 (in a freezer) at -80 °C for 2 h. The samples were then lyophilized in two freeze drying stages: 1) at -55 °C for  
141 1 h and 2) at 20 °C for 30 min. The second step prevents the formation of condensation when the samples  
142 are removed from the freeze dryer. Afterwards, the specimens were sputter coated with gold using an Agar  
143 automatic sputter coater, and a conductive bridge was “painted” between the sample top surface and the  
144 stub using Electrodag 1415, for those samples whose geometry prevented complete coating. ImageJ  
145 software was used to process images.

## 146 **2.8 Growth of *Lactobacillus acidophilus***

147 *L. acidophilus* was anaerobically propagated (100 µL) from a frozen stock in 10 mL MRS broth at 37 °C, for  
148 24 h and then 250 µL of the preculture was anaerobically incubated in 50 mL MRS broth at 37 °C, for 16 h in  
149 a glass bottle. After growth, the cells were harvested by centrifuging (4000 × g, 5 min, and 4 °C). After  
150 discarding the supernatant, the pellet was re-suspended in 0.1 M 50 mL phosphate buffered saline (PBS),  
151 pH 7.4 (8 g/L NaCl, 0.2 g/L, 1.44 g/L Na<sub>2</sub>HPO<sub>4</sub>, and 0.24 g/L KH<sub>2</sub>PO<sub>4</sub>) and centrifuged again. This washing  
152 procedure was repeated twice, and the cells were either incorporated in the hydrogels or re-suspended in  
153 50 mL PBS, pH 7.4 for enumeration.

## 154 **2.9 Incorporation of cells into hydrogels**

155 Initially, the CH (3 wt %) and DXS (2 wt %) stock solutions were sterilized by autoclaving at 121 °C for 20  
156 min. Before and after autoclaving the pH of the CH solution was approx. 4.8. The pH of DXS solution  
157 dropped from approx. 7.0 to 1.6 after autoclaving. However, the final pH of the mixture was approx. 4.4.  
158 The most promising hydrogel compositions were chosen for entrapment of cells, namely a physical

159 hydrogel containing 1.5 wt % CH and 1 wt % DXS, and a chemical crosslinked hydrogel containing 1.5 wt%  
160 CH, 1 wt % DXS, and 0.1 wt % GP. The procedure was as follows: 5 mL of CH (3 wt %) were introduced into  
161 the previously obtained washed cell pellet and vortexed briefly. 5 mL DXS (2 wt %) were added in the  
162 following step and vortexed again. For the formation of GP crosslinked gels, 10 mg of GP were added  
163 before incubating the gels at 37 °C and 225 rpm for 3 h.

#### 164 ***2.10 Viability and culturability of free and incorporated cells***

165 The viable cell count and culturability examination for free and incorporated bacteria into the hydrogels  
166 was done by the spread-plate method. After 3 h of incubation in the hydrogels, 40 mL PBS, pH 7.4 was  
167 added into 10 mL of the hydrogel and further homogenized in a stomacher (Seward, Stomacher 400, Lab  
168 System) using proper bags with side filter for 2 min at normal speed in order to release the cells trapped in  
169 the hydrogel. As a positive control, free cells were plated after being washed with PBS, pH 7.4. In the  
170 spread-plate method, 0.1 ml of an appropriately diluted culture using 0.1 M PBS, pH 7.4 was spread over  
171 the surface of previously prepared MRS agar plates in duplicates. The plates were then incubated  
172 aerobically at 37 °C for 48 h. The viability count for the suspension of free or released cells from hydrogels  
173 was expressed as colony forming units per milliliter (CFU/mL). The data are reported as the average of the  
174 number of viable cells  $\pm$  standard deviation from the duplicated experiments of duplicated plating.

175 Moreover, a qualitative viability assessment was carried out using the live/dead bacterial viability kit for  
176 1) the cells released from physical hydrogel, and 2) the residual part of the hydrogel (containing unreleased  
177 cells) in the stomacher bag after homogenization. It is essentially a two-colour fluorescence assay kit. The  
178 stain was prepared and applied following manufacturer's instructions. Photomicrographs of cells were  
179 acquired using Zeiss Axioskop microscope (Carl Zeiss, Goettingen, Germany) equipped with a Cool Snap RS  
180 Photometrics camera (Roper Scientific, Tucson, AZ, USA) and using an appropriate fluorescence filter. The  
181 images were taken with 40 x objectives and they were processed using the ImageJ software.

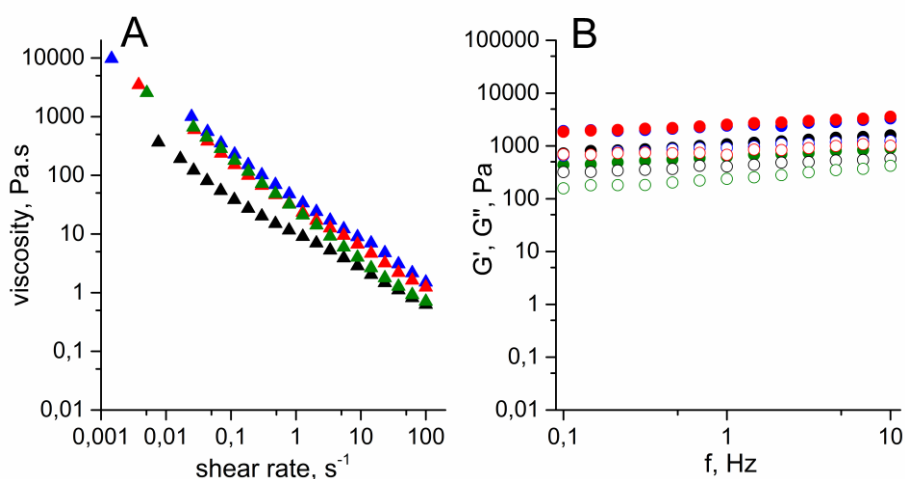
### 182 **3 Results and discussion**



### 183 3.1 *Gel formation and rheology*

184 The gelling process was slow and continuously developed throughout the 20-24 h of gentle stirring,  
185 which indicates a slow organization of the network. Fresh hydrogels were observed to be formed only in  
186 the CH and DXS concentration region of approx. 1-2 wt % and 1-2.5 wt %, respectively. However, with time  
187 and/or centrifugation it became clear that some of the formulations initially labeled as gels were not stable  
188 and eventually phase separated. The stable hydrogels were formed when 1.5 wt % CH was mixed with 1 or  
189 1.5 wt % DXS. In such systems, the gain in entropy upon mixing oppositely charged polyelectrolytes is due  
190 to release of the bound counterions that drives the associative phase separation (Kronberg, Holmberg, &  
191 Lindman, 2014). Hence, stable hydrogels were found to be formed at a charge density ratio ( $n^+/n^-$ ) of  
192 around 0.95-1.4 (see SI for charge density calculations). It is out of the scope of this paper to investigate in  
193 detail the phase behaviour of CH and DXS mixtures, but it is suggested that the observed phase separation  
194 for some formulations might be due to the expelling excess solvent after the formation of the coacervate.  
195 The reason behind such a phenomenon (also known as syneresis) is likely the thermodynamic imbalance,  
196 which leads to the reorganization of the polymer chains within the hydrogel network and consequent  
197 exclusion of solvent (Delmar & Bianco-Peled, 2015). This is a typical consequence of associative phase  
198 separation.

199 The samples that were visually identified as hydrogels were further examined regarding their rheological  
200 properties. Figure 1A shows typical flow curves for mixtures of CH and DXS exhibiting a non-Newtonian  
201 shear thinning behaviour regardless of the DXS concentration. The flow curves were also reversible with a  
202 small hysteresis area when shear rate was consecutively increased and decreased (data not shown). In  
203 Figure 1B, typical oscillatory frequency sweep tests are also depicted. In all the cases, the  $G'$  was higher  
204 than the  $G''$  and essentially frequency independent. These are the typical rheological fingerprints for gel-  
205 like materials (Picout & Ross-Murphy, 2003).



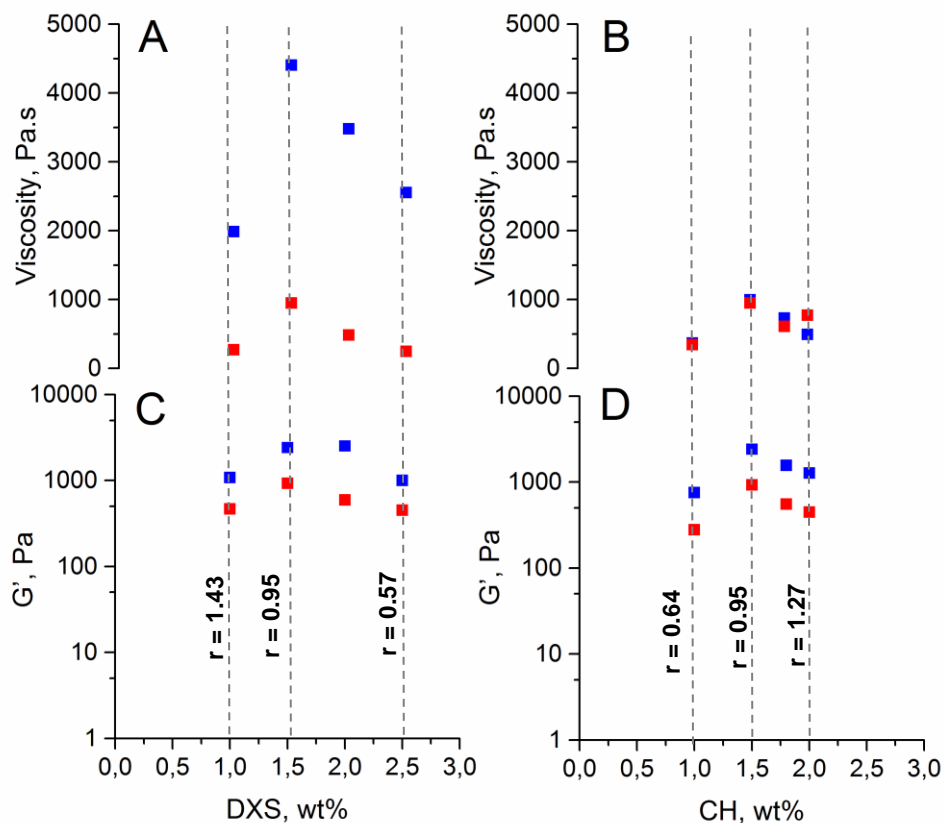
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207 Figure 1. Rheological properties of physical hydrogels: A) typical flow curves and B) typical dynamic  
 208 mechanical spectra (full symbols represent  $G'$  while empty symbols represent  $G''$ ) of preparations of 1.5 wt  
 209 % CH with varied DXS concentrations: 1 wt % (black), 1.5 wt % (blue), 2 wt % (red), and 2.5 wt % (green) at  
 210 25 °C.

211

212 Similar rheological profiles were observed for the other formulations and in Figure 2 the shear viscosity  
 213 (at  $\dot{\gamma} = 0.01 \text{ s}^{-1}$ ),  $G'$  and  $G''$  at a constant frequency are reported at 25 and 37 °C. The temperature effect is  
 214 clear; regardless of the CH and DXS concentrations, shear viscosity and  $G'$  were lower at 37 °C than at 25 °C  
 215 and this can be considered a normal temperature effect where, in general, polymer chains in solution  
 216 become more mobile with increased temperature. Another interesting feature is that both  $G'$  and shear  
 217 viscosity seemed to reach a maximum at a charge density ratio close to 1. This trend was observed while  
 218 keeping CH concentration constant and changing DXS concentration or vice-versa. Such an increase in the  
 219 rheological properties is explained by the maximized electrostatic interactions and physical entanglements  
 220 close to the 1:1 stoichiometry. However, when the entire polymer chains become neutralized, further  
 221 increase of one of the polymers does not improve the 3D structure. In fact, above the 1:1 ratio, the system  
 222 becomes overcharged, and polymer repulsion within the gel weakens the structure. This kind of complex

223 pattern was reported previously, in gels prepared with cationic cellulose derivative (cat-HEC) and double-  
 224 stranded DNA (dos Santos, Piculell, Medronho, Miguel, & Lindman, 2012).



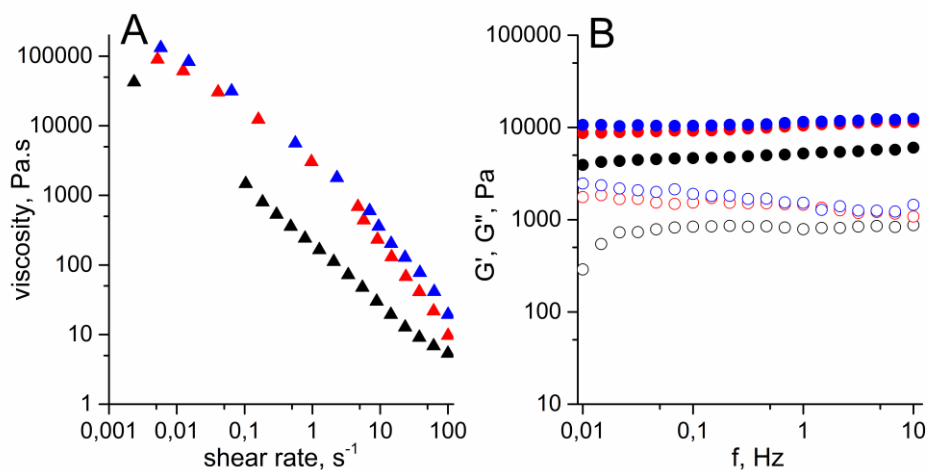
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226 Figure 2. Shear viscosity (at  $\dot{\gamma} = 0.01 \text{ s}^{-1}$ ) and  $G'$  (1 Hz) as a function of DXS (CH kept constant at 1.5 wt %)  
 227 and CH (DXS was kept constant at 1.5 wt %) concentrations at 25 °C (red) and 37 °C (blue). The pH of  
 228 mixtures was always between 4.7 - 4.9. Different charge ratios,  $r$ , are inserted as vertical dashed lines.

229

230 The most promising physical systems were selected to be crosslinked by GP with focus on the hydrogels  
 231 prepared with 1.5 wt % CH and 1 wt % DXS. Typically, hydrogels crosslinked with GP exhibited a dark blue  
 232 colour, which is associated with GP reaction with the amino groups of CH (Dimida et al., 2015). The  
 233 rheology tests were performed on the chemically crosslinked gels after about 2-3 days of crosslinking

234 reaction. In such conditions, GP is expected to have reacted and therefore the system should have reached  
 235 equilibrium. In Figure 3A, the chemical hydrogels showed a non-Newtonian shear thinning behaviour  
 236 regardless of GP concentration. As expected, the oscillatory frequency sweep showed a typical gel-like  
 237 pattern independent of GP concentration (Figure 3B).  
 238



239  
 240 Figure 3. Rheological properties of GP crosslinked hydrogels prepared with 1.5 wt % CH and 1 wt % DXS: A)  
 241 typical flow curves and B) typical dynamic mechanical spectra (full symbols represent  $G'$  while empty  
 242 symbols represent  $G''$ ) for GP concentrations of 0.05 (black), 0.1 (red) and 0.15 wt % (blue) at 25 °C.

243  
 244 Increasing the GP concentration led firstly to a pronounced increase on the viscosity and  $G'$  while no  
 245 major changes were observed above 0.1 - 0.15 wt % GP (see SI, Figure S1). Linear relations between the GP  
 246 content and the viscoelastic properties (i.e.  $G'$ ) was reported for CH chemical hydrogels (Dimida et al.,  
 247 2015; Moura, Figueiredo, & Gil, 2007). The plateau observed above 0.15 wt % GP might be related to the  
 248 saturation of CH with GP. The maximum in the rheological properties corresponds roughly to an estimated  
 249 stoichiometry of 1:6 (i.e. one genipin crosslinked to chitosan every 6 units). Since  $G'$  does not change upon  
 250 GP increase, this suggests the increase in crosslinking density is most likely hindered due to steric effects.  
 251 Moreover, the role of DXS in the 3D network can contribute to a more complex behaviour as compared to

252 simple CH systems. Particularly, the shear viscosity and  $G'$  increased about approx. 2-3 times, when DXS  
253 was added in the hydrogel in comparison with the systems without DXS (see SI, Table S2). Additionally, the  
254 GP effect is striking; the shear viscosity increased approx. 100 times while  $G'$  was enhanced approx. 7 times  
255 (see SI, Table S2). **One of the requirements for an efficient material to deliver probiotic cells is its capability**  
256 **to protect the cells from the imposed external mechanical stresses. Thus, the enhancement of the**  
257 **mechanical resistance of the matrix improves the conditions for a safe passage of the encapsulated cells**  
258 **through the gastrointestinal tract (Vos, Faas, Spasojevic, Sikkema, & De Vos, 2010).** Overall, the chemical  
259 gels are stronger and stiffer than the physical counterparts and they seem to display mechanical properties  
260 highly suitable for probiotic encapsulation and delivery.

### 261 **3.2 *Bead Formation***

262 So far the discussion has focused on bulk physical and chemical hydrogels. However, for several  
263 applications, the encapsulation of probiotics requires small delivery systems. Therefore, it is important to  
264 investigate whether the formulations screened for the bulk hydrogels can also be used to produce smaller  
265 particles. Different concentrations of CH and DXS were tried in order to form beads by either dripping CH  
266 into DXS or vice versa. In order to obtain physical beads, very high initial concentrations of DXS (approx.  $\geq$   
267 10 wt %) were needed (see SI, Figure S2A). On the other hand, the chemically crosslinked beads were  
268 formed by dripping CH into the solution of DXS and GP (see SI, Figure S2B). Crosslinked beads were  
269 successfully obtained by either dripping 2-3 wt % CH into a solution of 1-2 wt % DXS containing a fixed  
270 amount of GP (0.1 wt %). The dark blue colour of the beads is indicative of a successful crosslinking and was  
271 observed after allowing the reaction to run for 20 h at room temperature.

### 272 **3.3 *Hydrogel swelling***

273 The swelling kinetics is an important property for different applications of hydrogels. Therefore, the  
274 hydrogel composition (CH-DXS ratio) and crosslinker concentration in the hydrogel were investigated  
275 regarding their effect on the swelling behaviour of the systems at low pH, 2.0 and high pH, 7.4. The swelling

276 isotherms are represented in Figure 4 for pH 2.0 (for pH 7.4 see SI, Figure S3). A linear behaviour was  
277 observed in all the cases regardless of pH. Moreover, Figure 5 shows the variation of the equilibrium degree  
278 of swelling as a function of DXS (0, 1, 1.25 wt %) and GP (0, 0.05, 0.1 and 0.15 wt %) concentrations.  
279 Regardless of the pH, the equilibrium swelling decreased as the DXS concentration increased in the  
280 hydrogel (Figure 5A). This might be due to the physical reinforcement of the 3D network when the DXS  
281 approaches the charge ratio of CH, which is in agreement with the rheological data in Figure 2. Upon  
282 increasing DXS concentration (up to 1.5 wt %), the hydrogel became stronger (i.e. solid-like) and showed a  
283 more restricted swelling. Similar swelling profile in acidic (pH 2.0) and basic (pH 8.0) media was previously  
284 observed for hydrogel of CH (with a DA of 18.1 %) and DXS (Sakiyama, Takata, Kikuchi, & Nakanishi, 1999).  
285 Moreover, increasing the GP concentration (from 0 to 0.15 wt %) in the gel matrix led to a decrease in the  
286 equilibrium swelling percentage regardless of the pH (Figure 5B). Similar observations were reported, for  
287 instance, for hydrogels of dimethylaminoethyl acrylate methyl chloride quaternary salt crosslinked with  
288 acrylic acid (Katime, 2010). In general, the deformation of a hydrogel becomes more restricted by  
289 increasing the crosslinker content and often this leads to a less flexible hydrogel structure. Again, such  
290 swelling behaviour performance is in good agreement with the rheological properties of the chemical  
291 hydrogels presented in Figure S1, where the  $G'$  and shear viscosity are observed to increase with the GP  
292 concentration from 0 to 0.15 wt %. Note that, the physical gel (CH/DXS) dissolved and/or considerably  
293 degraded at pH 2.0. This result indicates the importance of crosslinking of the hydrogel to be able to stand  
294 the harsh acidic conditions of the stomach.

295 The swelling kinetics was investigated by fitting the experimental data assuming a second order kinetics.  
296 Since the water content at different times ( $W$ , %) can be expressed by

$$297 \quad W, \% = \frac{mt - m_0}{mt} \times 100 \quad (2)$$

298 Thus, the swelling rate at any time is expressed by (Katime, 2010; Schott, 1990)

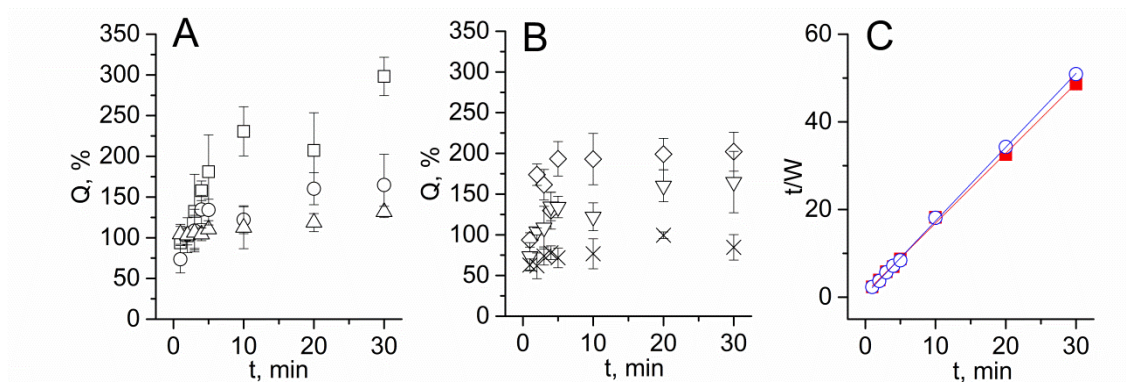
$$299 \quad \frac{dW}{dt} = K(W_{eq} - W)^2 \quad (3)$$

300 where  $W_{eq}$  is the water content at the equilibrium and  $K$  is kinetics rate constant.

301 Rearranging Eq. 3, it can be demonstrated that the data must fit a straight line with a slope of  $1/W_{eq}$  with  
302 ordinate of  $1/KW_{eq}^2$ . Then, the kinetics rate constant can be calculated from the following relation (Katime,  
303 2010):

$$304 \quad \frac{t}{W} = \frac{1}{KW_{eq}^2} + \frac{t}{W_{eq}} \quad (4)$$

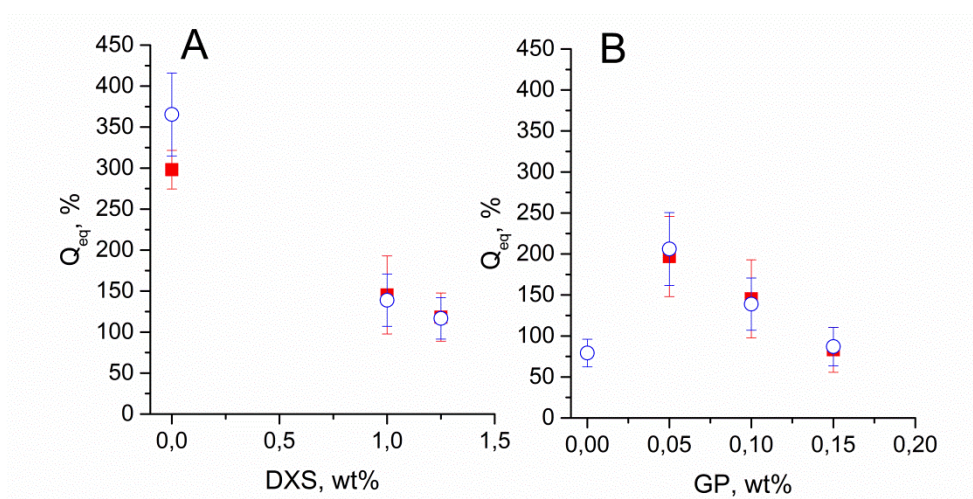
305 The experimental data were well fitted by Eq.4 with  $R^2 \geq 0.99$  for all the tested hydrogels as exemplified  
306 in Figure 4C. In general, all the analyzed crosslinked hydrogels achieved the swelling equilibrium reasonably  
307 fast (i.e. during the first 5 min), except for the chemical system without DXS. This fast swelling equilibrium  
308 was also observed in other systems such as for UV-crosslinked hydrogels of CH and (poly)acrylic acid (Lee et  
309 al., 1999). The extracted kinetic constants are summarized as a function of a) DXS and b) GP content (see SI,  
310 Table S3). It can be observed that increasing the DXS concentration led to an increase of the kinetic rate  
311 constant regardless of the pH. Nevertheless, the increase in the rate constant was more pronounced at pH  
312 7.4 than at pH 2.0. This is probably due to CH deprotonation at higher pH than the pKa of CH (approx. 6.5)  
313 that leads to weaker electrostatic interactions with the anionic DXS, and thus a more flexible porous  
314 structure capable of swelling faster and to a larger extent. On the other hand, increasing the GP  
315 concentration led to a decrease in the kinetic rate constant at pH 7.4 while the behaviour at low pH was not  
316 trivial. A very similar trend was found in a related system; Katime et al. reported an increase in the kinetic  
317 rate constants with dimethylaminoethyl acrylate methyl chloride quaternary salt (Q9) while such kinetic  
318 constants decreased with the crosslinker concentration for poly(acrylic acid-coQ9)-based hydrogels  
319 (Katime, 2010).



320

321 Figure 4. Swelling isotherms of hydrogels at pH 2.0 for compositions: A) 1.5 wt % CH and 0.1 wt % GP with  
 322 varied DXS (squares: 0 wt%, circles: 1 wt %, and triangle: 1.25 wt %), B) 1.5 wt % CH and 1 wt % DXS with  
 323 varied GP (diamonds: 0.05 wt%, inverse triangles: 0.1 wt %, and crosses: 0.15 wt %). C) Data and fits to Eq.  
 324 4 for hydrogels composed of 1.5 wt% CH, 1 wt% DXS and 0.1 wt% GP using Eq. 3 at pH 2.0 (red) and pH 7.4  
 325 (blue). Error bars represent the standard deviation from triplicate experiments.

326



327

328 Figure 5. Equilibrium swelling percentage ( $Q_{eq}$ , %) as a function of A) DXS concentration, and B) GP  
 329 concentration at pH 2.0 (red) and 7.4 (black). The hydrogels in A were prepared with fixed CH (1.5 wt %) and GP (0.1 wt %) concentrations while hydrogels in B were prepared with fixed CH (1.5 wt %) and DXS (1  
 330 wt %) concentrations. Error bars represent the standard deviation from triplicate experiments.

332

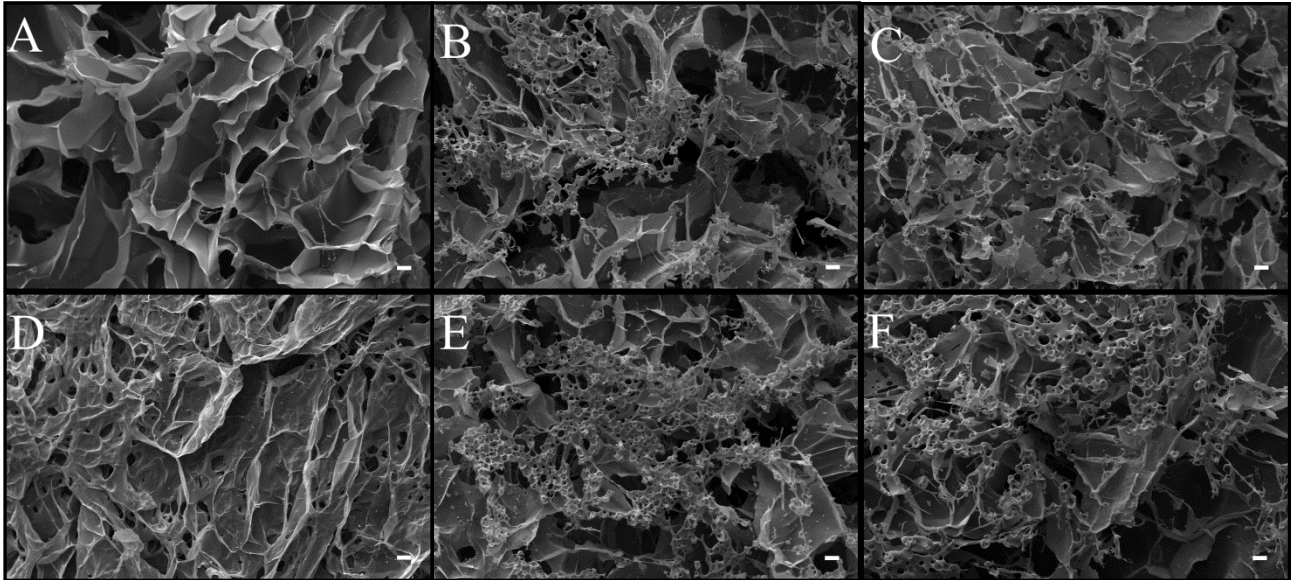


333 **3.4 Morphology of hydrogels**

334 The microstructure of chemical hydrogels was investigated by SEM imaging in order to determine the  
335 effect of DXS, CH, and GP contents on the hydrogel morphology. The chemical system without DXS (Figure  
336 6A) showed a highly porous structure with smooth surfaces. When DXS was progressively added (Figure 6B,  
337 C), the average size of the pores seemed to decrease, and the surfaces became rougher (see SI, Figure S4,  
338 for a higher magnification). Likewise, the degree of reticulation (smaller pores) and roughness of the  
339 surfaces increased with the GP concentration (Figure 6D, B, E). When the CH amount was increased, the  
340 same pore structure was observed, although the reticulation degree was lower since the CH/GP ratio was  
341 higher (see Figure 6C, F) Therefore, the proper combination of CH, DXS, and GP creates a more structured  
342 gel with a stronger network (higher reticulation and smaller pores), which is remarkably different from the  
343 morphology of the physical hydrogels or chemically crosslinked CH without DXS.

344 As discussed earlier, a possible use for the crosslinked hydrogels of CH and DXS is as carriers for probiotic  
345 delivery, and therefore the structure of crosslinked beads formed by the dropwise method was compared  
346 with the bulk hydrogels for the same composition (Figure 7). Essentially, the same microstructure and  
347 porosity were observed for both bulk and dropwise hydrogel beads.

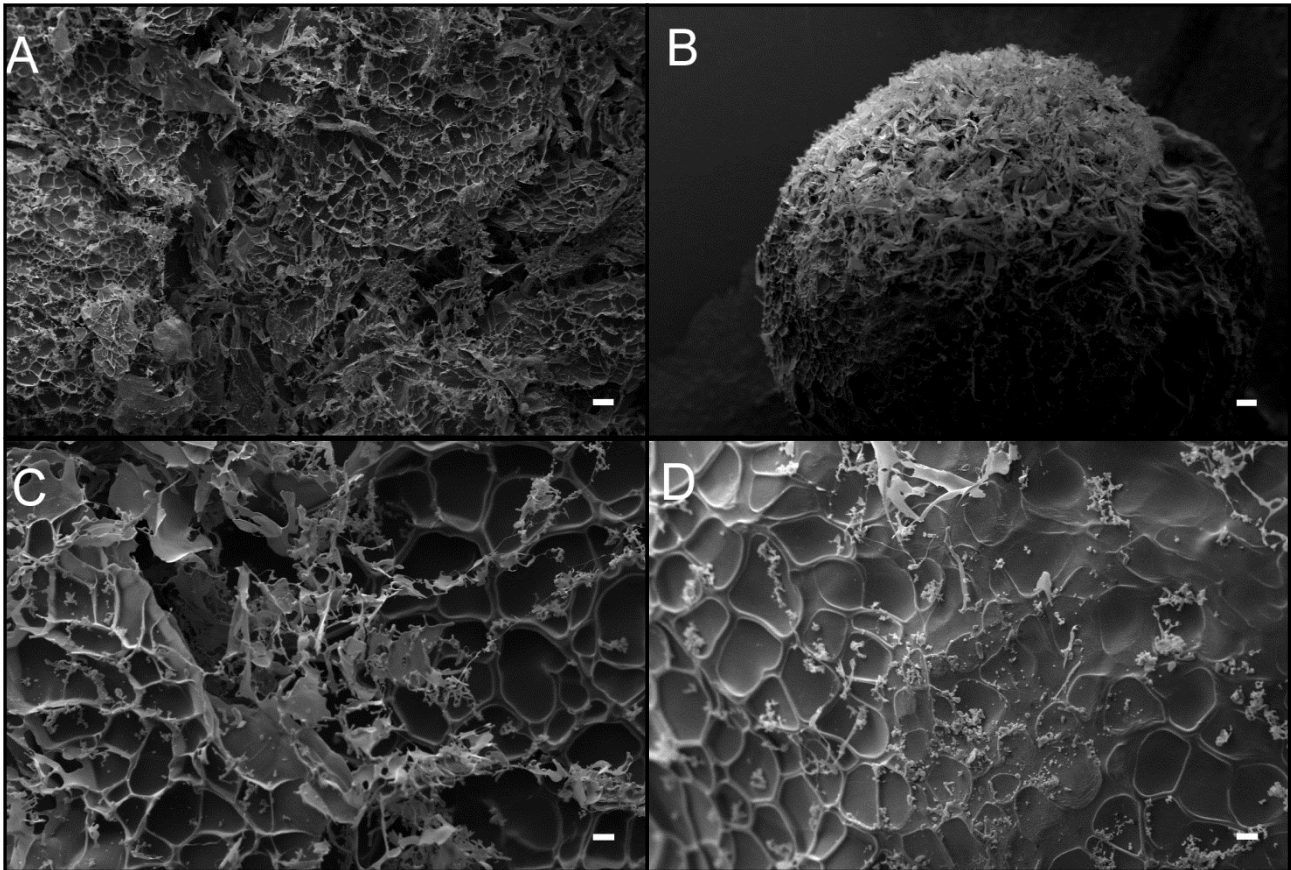
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349

350 Figure 6. SEM micrographs showing the effect of DXS (A, B and C), GP (D, B and E), and CH (C and F)  
351 concentrations on the morphology of hydrogels. The composition of hydrogels in the images are as follows:  
352 A) 1.5 wt% CH, 0.1 wt% GP, B) 1.5 wt% CH, 1 wt% DXS, 0.1 wt% GP, C) 1.5 wt% CH, 1.25 wt% DXS, 0.1 wt%  
353 GP, D) 1.5 wt% CH, 1 wt% DXS, E) 1.5 wt% CH, 1 wt% DXS, 0.15 wt% GP, and F) 2.1 wt% CH, 1.25 wt% DXS,  
354 0.1 wt% GP. The scale bar represents 20  $\mu\text{m}$ .

355



356

357 Figure 7. SEM micrographs of a bulk hydrogel (A and C) and a macro-bead (B and D) with the composition of  
 358 3 wt % CH, 2 wt % DXS and 0.1 wt % GP. The scale bars represent 100  $\mu\text{m}$  (top), and 20  $\mu\text{m}$  (bottom).

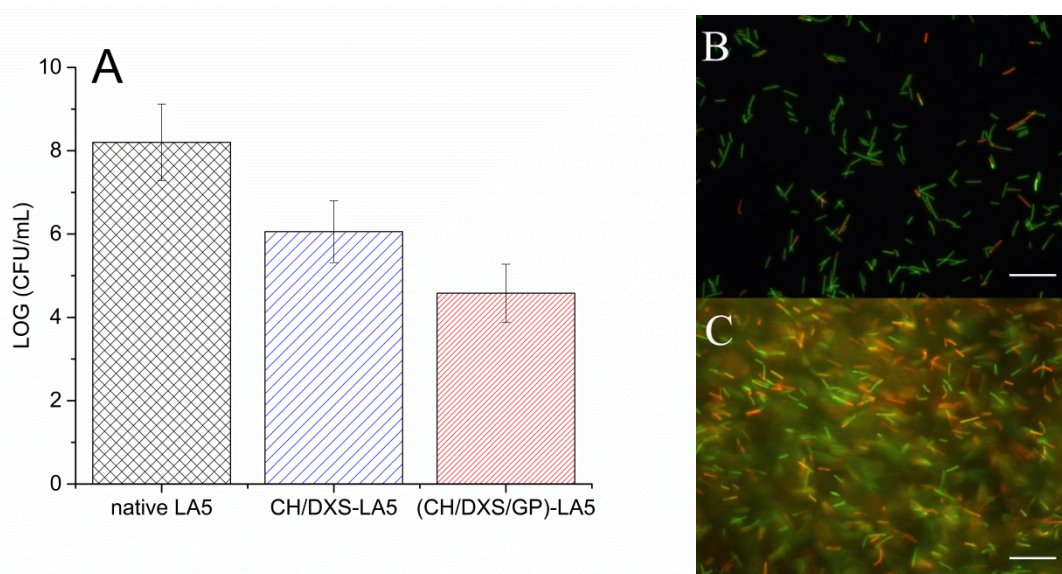
359

### 360 3.5 Cell viability and culturability

361 *L. acidophilus* (LA5) was the chosen model bacteria to be incorporated into the most promising physical  
 362 and chemical hydrogels. The goal was to show the ability of the cells to divide and form offspring **after**  
 363 **interacting with different hydrogel matrices**. The LA5 cell culturability was tested on solid agar surfaces and  
 364 compared with the native free cells. The number of viable native cells was  $8.2 \pm 0.9$  log CFU/mL. Figure 8A  
 365 shows approx. 2.1 log CFU/mL and approx. 3.6 log CFU/mL decrease for the cells entrapped in the physical  
 366 and chemical hydrogels, respectively, when compared to the number of viable native bacteria. A qualitative  
 367 assessment of the viability was also performed using a live/dead staining kit (Figure 8). In Figure 8B, approx.  
 368 70-80 % of cells seem to be alive (green) after incorporation into the physical gel. This is in good agreement

369 with the plate counting results. The reason behind the decrease in the number of viable cells due to  
370 entrapment in hydrogels might be in part due to the hydrogel composition (if it is not neutralized, CH can  
371 display antimicrobial properties against some strains) or in part due to the rather low pH of the hydrogel  
372 (approx. 4.4). Additionally, the cell counting might be underestimated since some unreleased cells may  
373 have remained trapped in the hydrogel matrix (a small gel fraction remained as residue in the stomacher  
374 bag).

375 In the case of the chemical systems, the number of viable cells was less ( $4.6 \pm 0.7 \log \text{cfu/mL}$ ) than the  
376 number of viable native cells. However, a significant amount of gel residue remained in the stomacher bag  
377 after homogenization in this case. Since this residue was not accessible, it could not be plated and counted  
378 (Figure 8C). Moreover, at least approx. 50 % of the cells in residue were estimated to be alive using  
379 live/dead viability kit in this case. Hence, bacterial cells might be viable, but not culturable. Finally, it is  
380 possible, however, that the chemical hydrogels could be further degraded in the gastrointestinal tract due  
381 to enzymatic action of gastrointestinal bacterial enzymes (Kean & Thanou, 2010).



382  
383 Figure 8. Viability of *L. acidophilus* (LA5) in physical hydrogel (1.5 wt % CH, 1 wt % DXS) and GP crosslinked  
384 hydrogel (1.5 wt% CH, 1wt% DXS, 0.1 wt% GP): A) plate counting results, B) microscopy image showing the  
385 live and dead cells released from a physical hydrogel and C) microscopy image showing the residue of the  
386 chemical gel exposed to live/dead staining viability kit. The scale bar represents 20  $\mu\text{m}$ .

387

#### 388 **4 Conclusions**

389 In this work, physical and GP crosslinked hydrogels of CH and DXS were developed and characterized as a  
390 potential delivery system material for probiotic bacteria encapsulation targeted the gastrointestinal tract.  
391 Physical and chemical hydrogel formulations were identified with a general non-Newtonian shear thinning  
392 and gel-like behaviour regardless of DXS, CH, and GP concentrations. An optimum CH / DXS ratio was  
393 determined (approx. 1:1) where the rheological properties were maximized independently of the pH and  
394 temperature. However, the chemical hydrogels (crosslinked with GP) were more robust and stiffer than the  
395 related physical hydrogels. The swelling of the chemical hydrogels was found to be strongly dependent on  
396 DXS and GP concentrations, not influenced by the pH of the media and obeyed a second order kinetics  
397 swelling profile. Generally, increasing either the DXS or GP led to a decrease in equilibrium swelling that  
398 was found to be in good agreement with the hydrogels microstructure (i.e. smaller pores and higher  
399 reticulation degree for samples with higher amounts of DXS and GP). Apart from the bulk hydrogels, beads  
400 were formed by the dropwise method and found to be remarkably similar to the related bulk hydrogels in  
401 terms of morphological properties. Finally, the viability and culturability tests revealed approx. 2.1 and 3.6  
402 log CFU/mL decrease in the number of viable cells, which were incorporated in physical and chemical  
403 hydrogels, respectively. However, such values are underestimated since the qualitative viability assay  
404 (live/dead viability kit) evidenced that a considerable amount of viable cells were still entrapped in the  
405 hydrogel matrix and not released in the media used for plating. **The improved mechanical properties might  
406 lead such insufficient release behavior in the case of the chemical hydrogels. It remains to be studied if the  
407 presence of specific enzymes, bile salts or other molecules can improve the release of La5 from such robust  
408 gel matrixes.** Overall the systems developed are highly flexible in terms of structure and mechanical  
409 properties. By fine control of the composition and crosslinking degree it is possible to tune different  
410 features of interest such as viscoelasticity, gelling ability, swelling and microstructure/morphology.  
411 Therefore, the systems developed in this work are promising matrices for probiotic cell encapsulation and

412 delivery. In the future work, high focus will be given to the probiotic cells encapsulation in hydrogel beads  
413 and the characterization of such systems in terms of viability of the encapsulated cells, their survivability in  
414 simulated gastrointestinal fluid, and their release in simulated intestinal fluid.

#### 415 **Acknowledgements**

416 The research leading to these results has received funding from the People Programme (Marie Curie  
417 Actions) of the European Union's Seventh Framework Programme FP7/2007-2013/ under REA grant  
418 agreement n° 606713. Bruno Medronho acknowledges the Portuguese Foundation for Science and  
419 Technology (FCT) through the project PTDC/AGR-TEC/4814/2014 and researcher grant IF/01005/2014.  
420 Marité Cárdenas thanks the Swedish Research Council for financial support, and acknowledges the  
421 umbrella of COST actions CM1101 and MP1106.

#### 422 **Appendix A. Supporting Information**

423 Supplementary data associated with this article can be found, in the online version, at < insert the link >.

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